Nanodomain Engineering for Programmable Ferroelectric Devices

Alexey Lipatov,†‡ Tao Li,‡,# Nataliia S. Vorobeva,‡ Alexander Sinitskii,*,†,# and Alexei Gruverman*,‡,#§

†Department of Chemistry, University of Nebraska, Lincoln, Nebraska 68588, United States
‡Department of Physics and Astronomy, University of Nebraska, Lincoln, Nebraska 68588, United States
#Centre for Spintronics and Quantum System, State Key Laboratory for Mechanical Behavior of Materials, School of Materials Science and Engineering, Xi’an Jiaotong University, Xi’an, Shaanxi 710049, People’s Republic of China
§Nebraska Center for Materials and Nanoscience, University of Nebraska, Lincoln, Nebraska 68588, United States

Supporting Information

ABSTRACT: We introduce a concept of programmable ferroelectric devices composed of two-dimensional (2D) and ferroelectric (FE) materials. It enables precise modulation of the in-plane conductivity of a 2D channel material through nanoengineering FE domains with out-of-plane polarization. The functionality of these new devices has been demonstrated using field-effect transistors (FETs) fabricated with monolayer molybdenum disulfide (MoS2) channels on the Pb(Zr,Ti)O3 (PZT) substrates. Using piezoresponse force microscopy (PFM), we show that local switching of FE polarization by a conductive probe can be used to tune the conductivity of the MoS2 channel. Specifically, patterning of the nanoscale domains with downward polarization creates conductive paths in a resistive MoS2 channel so that the conductivity of an FET is determined by the number and length of the paths connecting source and drain electrodes. In addition to the device programmability, we demonstrate the device ON/OFF cyclic endurance by successive writing and erasing of conductive paths in a MoS2 channel. These findings may inspire the development of advanced energy-efficient programmable synaptic devices based on a combination of 2D and FE materials.

KEYWORDS: Molybdenum disulfide, lead zirconium titanate, ferroelectric memory, field-effect transistor (FET), domain patterning

In recent years, there has been a considerable interest in the hybrid electronic structures comprising two-dimensional (2D) and ferroelectric (FE) materials.1−11 These structures possess appealing characteristics for application in nonvolatile memory and logic devices and may offer low-power, energy-efficient alternatives to CMOS processing. One of such attractive characteristics is memristive functionality, electrically tunable resistivity, which can be used for emulating synaptic functions in neuromorphic computing applications.12−27 The memristive devices are typically two-terminal structures, in which a layer of a switchable material is sandwiched between two electrodes. In this Letter, we demonstrate a new concept of an electronic device with programmable resistivity based on 2D and FE materials.

The 2D-FE devices can be divided in two general groups based on their architecture and functionality. The first group is ferroelectric field-effect transistors (FeFETs) (Figure 1a), where the in-plane conductivity of a 2D channel material is modulated by the polarization of an FE gate dielectric.9−11 The second group includes ferroelectric tunnel junctions (FTJs), devices comprising a nanometer-thick FE layer sandwiched between two electrodes (one of which is a 2D material),12,13 in which the perpendicular-to-plane tunneling conductance can be tuned by switching the ferroelectric polarization (Figure 1b). The original devices of these kinds employed graphene as a channel material (in FeFETs) or a top electrode (in FTJs) and exhibited low ON/OFF ratios due to the semimetallic nature of graphene.8,14−19 Later on, it was shown that the ON/OFF ratios could be significantly enhanced if graphene was replaced with a semiconducting 2D material with a substantial band gap, such as molybdenum disulfide (MoS2).13,20−25 For example, FeFETs employing a MoS2 device channel on Pb(Zr,Ti)O3 (PZT) exhibit a nonvolatile ON/OFF ratio of over 100, an order of magnitude improvement compared to similar FeFETs based on graphene.20 In the case of FTJs with the MoS2 top electrodes, the ON/OFF ratio enhancement compared to their graphene-based counterparts is even more dramatic, by almost 2 orders of magnitude. In addition, it was reported that the coupling between ferroelectric polarization and conductivity of MoS2 resulted in an asymmetric and spatially confined switching behavior in MoS2−BaTiO3 FTJs, which allowed fine-tuning of the resistive states achieved via generation of stable sequential domain patterns in BaTiO3

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(BTO) (Figure 1b), adding memristive functionality to the FTJ devices. Here, we extend the concept of nanoscale domain engineering for resistance control to the hybrid 2D-FE FET devices (Figure 1c). Using the scanning probe microscopy (SPM) approach, we demonstrate a precise reversible modulation of the in-plane electronic transport in the MoS2−PZT FETs by creating conductive domain paths of a specific geometry.

While Figure 1c demonstrates a general concept of a programmable MoS2−FE device, below we provide a detailed description of nanodomain engineering by an electrical bias applied to the MoS2 layer via a conductive SPM tip as a way to control the device conducting state. This approach makes use of the asymmetric nature of polarization reversal in the MoS2−FE structures and the dependence of the MoS2 electronic properties on the FE polarization direction. Specifically, it has been shown that the downward (toward the bottom electrode) FE polarization drives a MoS2 flake into an electrically conductive state, which allows complete switching of polarization from the downward (P_{down}) to the upward (P_{up}) state in the whole volume of an FE material underneath MoS2 upon application of a single negative pulse from an SPM tip (Figure 1b). In turn, the upward FE polarization makes MoS2 more insulating so that application of a positive pulse...
results only in local switching of the polarization to the downward state right under the tip. This behavior is illustrated in Figure 2, where the N-shaped downward-oriented domain was created in the MoS₂/BTO FTJ by scanning with the tip under a positive bias of 4 V. Comparison of the domain pattern visualized using piezoresponse force microscopy (PFM) (Figure 2a,b) and a corresponding conducting atomic force microscopy (CAFM) map (Figure 2c) shows that BTO polarization strongly affects the out-of-plane conductivity of the MoS₂—BTO heterostructure: a much higher current (by almost 2 orders of magnitude) is detected in the region with the downward polarization \( P_{\text{down}} \). Kelvin probe force microscopy (KPFM) measurements reveal the corresponding variations in the contact potential difference (CPD) of MoS₂ caused by the charge accumulation/depletion in MoS₂ by the opposite polarization charges of the underneath BTO film (Figure 2d). Cross-sectional analysis of the PFM, CAFM, and KPFM maps (Figure 2e) shows a clear correlation between the polarization, local conductivity, and surface potential in the MoS₂—BTO heterostructure.

The same nanodomain engineering approach can be applied to modulate the in-plane conductivity of the MoS₂—PZT FETs (Figure 1a) with a single-crystalline monolayer MoS₂ channel bridging source (S) and drain (D) electrodes (Figure 3a). Polycrystalline 100 nm-thick (001)-oriented tetragonal PbZr₀.4Ti₀.6O₃ ferroelectric films used in this study were grown by metal—organic chemical vapor deposition (CVD) on the Ir/TiO₂/SiO₂/Si substrate. The CVD-grown monolayer MoS₂ flakes, characterized by Raman spectroscopy, were placed onto the PZT surface using the wet transfer technique. The Cr/Au source and drain electrodes contacting MoS₂ flakes were fabricated by the standard electron beam lithography and electron beam evaporation. Details of the experimental methods and device fabrication are given in the Supporting Information.

Figure 3b,c shows PFM images of the MoS₂—PZT device. The PZT film underneath the MoS₂ flake initially polarized downward was switched to the upward direction by the application of a −5 V pulse to the flake. The upward PZT polarization yields the OFF (low conductance) state of the PZT device with the drain—source current \( (I_{\text{DS}}) \) at the level of 3.0 nA at \( V_{\text{DS}} = 100 \text{ mV} \) (Figure 3f). Scanning a region between the planar source/drain electrodes with the tip under a +5 V dc bias at 500 nm/s results in a 200 nm-wide linear domain with the downward polarization (Figure 3d,e). The associated change in local conductivity of MoS₂ results in the appearance of a conducting channel between the source and drain electrodes, leading to an increase of \( I_{\text{DS}} \) up to 12.3 nA at \( V_{\text{DS}} = 100 \text{ mV} \) (see the red \( I_{\text{DS}}—V_{\text{DS}} \) dependence in Figure 3f), thereby driving the device to the ON state. The written domain width strongly depends on the writing bias but shows a relatively weak dependence on the tip scanning speed during writing (see Figures S1 and S2). This conductive path can be severed by applying a short negative pulse and then restored again by scanning with a positively biased SPM tip (see Figure S1). Erasure occurs when a negative pulse of −5 V is applied to the MoS₂ flake, switching the polarization of the entire PZT volume under the flake to the upward state.

Endurance of conductance modulation via the domain engineering has been tested by performing several cycles of writing and complete erasing of a domain line (Figure 3f). The high-resistance state exhibits \( I_{\text{DS}} \) in the range of 1–3 nA at \( V_{\text{DS}} = 100 \text{ mV} \), while for the low-resistance state with a single domain line the \( I_{\text{DS}} \) is in the range of 12–14 nA at \( V_{\text{DS}} = 100 \text{ mV} \) (see the inset in Figure 3f). The \( I_{\text{DS}}—V_{\text{DS}} \) curves for the OFF and ON states are reproducible from cycle to cycle. This experiment proves the reliability and programmability of the nanodomain engineering method as a resistance-modulation procedure for the hybrid 2D-FE devices.
PFM allows patterning arbitrarily shaped conducting paths, as illustrated by Figure S3, where a number of downward polarized domain lines were created in the channel of the MoS$_2$−PZT FET. Thus, nanoscale domain engineering allows for fine-tunability of resistance in the MoS$_2$−PZT FETs. Figure 4a shows PFM images recorded after creating several downward polarized domain lines by the SPM tip scanning under the positive bias of 5 V. The in-plane conductivity measurements, which were performed after writing each $P_{\text{down}}$ domain line, reveal the $I_{\text{DS}}$ currents of 6.5, 75, and 136 nA at $V_{\text{DS}} = 100$ mV for one, two, and three domain lines, respectively (Figure 4b). The device conductivity increases linearly with the number of lines as shown in Figure 4c, yielding the maximum ON/OFF ratio of about 130. The estimated ON/OFF ratio for the fully switched PZT underneath the MoS$_2$ flake is of the order of 1500.

In summary, we demonstrated the concept of a new programmable semiconductor device with a 2D channel and ferroelectric gating that enables precise modulation of the in-plane conductivity through the nanoengineering of ferroelectric domains with out-of-plane polarization. Functionality of these devices was demonstrated using MoS$_2$-based FETs fabricated on the PZT substrates. Anisotropic switching behavior of the MoS$_2$−PZT heterostructures enables the PFM patterning of arbitrarily shaped nanoscale downward domains, allowing local control of the MoS$_2$ conductivity. It is shown that generation of a conductive path in the MoS$_2$ layer connecting the source and drain electrodes by polarization switching in the predetermined regions of the underlying PZT film drives the MoS$_2$−PZT FET device from an OFF resistive state to an ON resistive state. The cyclic endurance and reproducibility of the resistance switching have been demonstrated by successive writing and erasing of the conductive paths in the MoS$_2$ channel. The ON state conductivity can be further modulated by creating a different number of conducting paths of variable geometry (i.e., shape and width) using a proper combination of the electrical bias and scanning trajectories, which opens a possibility for the development of novel synapse devices with multiple resistive states. Furthermore, the experimentally demonstrated programmability of the MoS$_2$−PZT FETs may inspire the development of innovative designs for energy-efficient synaptic devices.

**ASSOCIATED CONTENT**

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.nanolett.9b00673.

Materials and methods, images and analysis of CVD-grown MoS$_2$, and additional PFM images for different domain patterning parameters (PDF)

**AUTHOR INFORMATION**

Corresponding Authors

*E-mail: sinitskii@unl.edu.
*E-mail: alexei_gruverman@unl.edu.

ORCID

Alexey Lipatov: 0000-0001-5043-1616
Alexander Sinitskii: 0000-0002-8688-3451
Alexei Gruverman: 0000-0003-0492-2750

Author Contributions

A.L. and T.L. contributed equally. All authors have given approval to the final version of the manuscript.

Notes

The authors declare no competing financial interest.
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